

The Crystal Structure of Eu_3O_4 *

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The crystal structure of trieuropium tetroxide, Eu_3O_4 , has been determined from single-crystal X-ray diffraction data. This compound, containing both divalent and trivalent europium ions, is isomorphous with CaFe_2O_4 . It crystallizes in the orthorhombic system, space group *Pnam*. The unit cell has dimensions $a = 10.085$, $b = 12.054$, $c = 3.502$ Å, and contains four formula units. The structure is layered along the short *c* axis, having all atoms on mirror planes at $z = \frac{1}{4}$ and $z = \frac{3}{4}$. Least-squares refinement of *x* and *y* positional parameters and isotropic temperature factors was based on 955 observed independent *hk0* and *hk1* reflections, and established the structure with a final *R* of 0.089. The coordination of oxygens around trivalent europium is sixfold, forming a distorted octahedron around each Eu^{3+} ion. Around divalent europium the coordination is eightfold, with each Eu^{2+} ion lying at the center of a triangular prism of six oxygen atoms. Two oxygen atoms lying out from the centers of two of the prism faces complete the coordination.

Introduction

In the search for more efficient nuclear reactor control materials, ceramic compounds containing some of the rare earth ions having high neutron capture cross sections have been studied under conditions simulating possible reactor environments. During high-temperature tests of monoclinic europium sesquioxide (Eu_2O_3) under reducing conditions, it was found that this normally yellowish-white colored oxide could be reduced to a dark red-brown to blackish material. Powder X-ray diffraction patterns showed that this material was not the NaCl-type cubic divalent monoxide, EuO , reported by Eick, Baenziger & Eyring (1956), but indicated a more complex structure. Subsequent single-crystal studies indicated that this new compound was orthorhombic, having approximate unit-cell dimensions of $a = 10.07$, $b = 12.01$, and $c = 3.50$ Å, and had a measured density of 8.03 g.cm^{-3} . By comparison of this measured density with that calculated with various possible combinations of europium and oxygen in the unit cell, the compound was deduced to be trieuropium tetroxide, Eu_3O_4 , containing both divalent and trivalent cations.

The existence of a compound corresponding to the composition Eu_3O_4 was first reported by Achard (1960) He obtained an intermediate phase, having an oxygen to europium ratio of approximately 1.3, during his studies of carbon reduction of Eu_2O_3 to EuO . Later, Bärnighausen & Brauer (1962) reported the preparation of Eu_3O_4 by heating an equimolar mixture of EuO and Eu_2O_3 at 900°C in an inert atmosphere. Although these authors reported the lattice parameters and space group for Eu_3O_4 and the isomorphous compound SrEu_2O_4 , and showed that these compounds are structurally related to calcium ferrite, CaFe_2O_4 (Decker &

Kasper, 1957), they did not carry out the complete crystal structure analysis. Recently, Schieber & Holmes (1964) prepared mixtures of Eu_3O_4 in Eu_2O_3 during attempts to reduce the sesquioxide in carbon monoxide atmosphere.

Preparation of Eu_3O_4

For the present study, Eu_3O_4 was prepared by several methods. The first preparation occurred accidentally when Eu_3O_4 was formed as a granular black layer approximately one millimeter thick surrounding a core of Eu_2O_3 which had been hot-pressed in a graphite die at 3.5 kg.cm^{-2} (50 lb.in^{-2}) at 1700°C . Single-crystal fragments chipped from this layer were used for the present structure analysis.

Subsequent to this initial preparation, a reliable method of producing single-phase Eu_3O_4 was developed consisting of hydrogen reduction of the sesquioxide or the hydroxide. This method, which involved heating Eu_2O_3 or $\text{Eu}(\text{OH})_3$ in a hydrogen atmosphere at 1650°C , yielded coarsely polycrystalline black chunks of Eu_3O_4 . Single-crystal fragments from these chunks appeared dark red under the polarizing microscope, and were found to be biaxial with refractive indices near 2.12. Grinding chunks of Eu_3O_4 produced a red-brown powder.

Preliminary analyses

Preliminary Weissenberg and precession photographs established the orthorhombic symmetry, and showed reflections present for the following conditions:

$$h00, 0k0, 00l \text{ only when } h, k, l = 2n$$

$$h0l \text{ only when } h = 2n$$

$$0kl \text{ only when } k + l = 2n$$

$$hk0, hkl \text{ without regular extinctions.}$$

These conditions are satisfied by space groups *Pna*₂ (No. 33) and *Pnam* (*Pnma*, No. 62). Assignment of the

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centrosymmetric space group was based upon consideration of the reflection intensities. As pointed out by Bärnighausen & Brauer (1962), alternate layers along the short c axis exhibit similar intensities, thus indicating that all atoms must lie on two planes separated by $\frac{1}{2}$ in z . This condition is fulfilled only by the special fourfold positions (c) in the space group $Pnam$, where all atoms have $z = \frac{1}{4}$ or $\frac{3}{4}$.

Lattice parameters derived from the single-crystal films were refined by means of powder data. The Eu_3O_4 powder pattern was obtained with iron $K\alpha$ radiation on a Norelco diffractometer (Rau, 1964). Indexed back-reflection lines from that pattern were used to calculate the lattice parameters, using the computer program of Mueller, Heaton & Miller (1960). The values obtained are listed in Table 1, and are seen to agree closely with those reported by Bärnighausen & Brauer (1962). With these values, and assuming four formula units per unit cell, the theoretical density of Eu_3O_4 is 8.11 g.cm^{-3} , in good agreement with the measured value of 8.03 g.cm^{-3} .

Table 1. Lattice parameters of Eu_3O_4

	This study	Bärnighausen & Brauer (1962)
a	$10.085 \pm 0.002 \text{ \AA}$	$10.094 \pm 0.003 \text{ \AA}$
b	12.054 ± 0.002	12.068 ± 0.003
c	3.502 ± 0.001	3.500 ± 0.001

Computing procedures

For the structure analysis of Eu_3O_4 , the 'X-ray 63' crystal structure calculation system of Stewart & High (1964) was used on an IBM 7094 computer. This system is composed of an integrated set of computer programs for preparing the data and making the many different calculations involved in crystal structure analysis. Interconnection of the links of the system makes it possible to use the programs individually or in continuous sequence, with the output of one program serving as input to the next.

Collection of data

Although a preliminary set of Weissenberg photographs was obtained using copper radiation, these photographs were felt to be unsuitable for the structure analysis, since the extremely high absorption coefficient ($\mu = 3033.95 \text{ cm}^{-1}$) of Eu_3O_4 for Cu $K\alpha$ radiation would cause large errors in measured intensities. Accordingly, Mo $K\alpha$ radiation was chosen ($\mu = 438.75 \text{ cm}^{-1}$). The crystal used for intensity collection was approximately needle-shaped, being elongated along the c axis. Its cross-section was nearly square, with an average radius of 0.005 cm .

Equi-inclination Weissenberg photographs were taken of the ($hk0$) and ($hk1$) levels. Since the preliminary analysis had indicated a layered structure with all atoms on mirror planes at $z = \frac{1}{4}$ and $z = \frac{3}{4}$, no z parameter refinement was planned, and additional level data were not collected.

Intensities were collected with the use of the multiple film technique. Two film packs, composed of three sheets of Kodak No-Screen film separated by nickel foil, were exposed for 48 hours and 3 hours, respectively, for each level of data. Intensities were estimated visually by comparison with a calibrated intensity strip prepared using the 920 reflection of the crystal under investigation. All zones of the Weissenberg films were read, and intensities of crystallographically equivalent reflections were averaged. In all, 955 of a possible 1463 crystallographically different reflections had measurable intensities.

The averaged intensities were then correlated and scaled to the most heavily exposed film. For ($hk0$) and ($hk1$) films the average inter-film correlation factors within the film packs were 2.75 and 2.68, respectively, and the inter-pack factors were 14.68 and 13.80, respectively. The intensities were converted to experimental structure factors, F_o , in the usual way, with corrections being applied for the Lorentz and polarization effects, as well as for sample absorption. The absorption corrections used were based on a cylindrical crystal having $\mu R = 2.2$ (Bond, 1959).

The trial structure

In their work on Eu_3O_4 , Bärnighausen & Brauer (1962) found that the strong intensities showed general agreement with the strong reflections of calcium ferrite, CaFe_2O_4 (Decker & Kasper, 1957), thus indicating that the heavy atoms in Eu_3O_4 were probably located in positions similar to those occupied by the metal atoms in the reported structure of that compound. In the present study, good intensity agreement was found not only between the strong reflections, but also between many of the weaker reflections as well. Based on this agreement, a trial Eu_3O_4 structure containing trivalent europium ions in the iron positions and divalent europium ions in the calcium positions of CaFe_2O_4 was proposed.

To check this trial structure, calculations of structure factors and Fourier summations were made with the unit cell containing only europium atoms, using an over-all isotropic temperature factor of $B = 0.55 \text{ \AA}^2$ and the scattering factors of Cromer & Waber (1965). Fourier maps appeared reasonable, and sections at $z = \frac{1}{4}$ and $\frac{3}{4}$ showed not only the three strong europium peaks of the asymmetric unit, but also showed evidence of four weak peaks at locations equivalent to oxygen positions in CaFe_2O_4 . One anomalous intensity spike was present at a location related to that of the $\text{Eu}(3)$ atom by a translation of $y \pm \frac{1}{2}$, but this was removed by subsequent refinement. The agreement index at this time, calculated with no oxygen atoms in the cell, was $R = 0.34$.

The refinement

Refinement was carried out with the X-ray 63 system version of the Busing, Martin & Levy (1962) full-matrix

least-squares program, ORFLS. The function minimized was $\sum w(F_o - F_c)^2$, with $w=50/F_o$ when $F_o > 50$ and $w=1$ when $F_o \leq 50$. Unobserved reflections were entered as the minimum observable threshold values, and were weighted $w=1$ when $F_c > F_{\min}$ and $w=0$ when $F_c < F_{\min}$. Refinement was carried out on 23 variables, including the x , y , and B parameters for each of the seven atoms in the asymmetric unit, and two scale factors for the two levels of data. Starting x and y positional parameters were derived from the Fourier map, and individual isotropic temperature factors $B=0.55 \text{ \AA}^2$ were assigned to each atom. Ionic scattering factors were used, with those for Eu^{2+} and Eu^{3+} being taken from Cromer & Waber (1965), and those for O^{2-} being taken from Suzuki (1960).

The R value of the observed reflections with oxygen ions inserted in the lattice was 0.325; this was reduced to 0.195 after two least-squares cycles. Shifts in all parameters were appreciable, especially for the oxygen temperature factors. A Fourier map computed at this time showed sharpening of the electron density peaks, and reduction of the anomalous intensity spike to a negligible value.

Eight additional least-squares cycles produced convergence to an R of 0.095. A Fourier map computed at this time was quite similar to the one computed with $R=0.195$, except that the anomalous spike was completely absent.

At this point, it was noticed that low-angle reflections with $(\sin \theta)/\lambda < 0.30$ were systematically stronger than their calculated F_c values. Inspection of the Weissenberg films showed this to be related to a noticeable background which fell off abruptly at this angle. Accordingly, the 41 observed reflections within the affected range were given an empirical correction derived from a plot of their F_c/F_o ratios. Also at this time, nine reflections which were judged to be affected by secondary extinction were given zero weight. In addition, the 400 and 240 reflections, which lay on strong white radiation streaks, were also weighted zero. Two additional cycles of least-squares with the corrected data produced only slight changes in the parameters, and lowered R to 0.089.

The final parameters derived from the last least-squares cycle are listed in Table 2, together with their errors, which were derived from the diagonal elements of the inverse least-squares matrix. All parameter shifts during the last least-squares cycle were less than 0.12σ , the average shift/error ratio being 0.039. The observed

structure factors are compared with those calculated from the final parameters in Table 3. The final R , omitting unobserved reflections, was 0.089, while the weighted R was 0.119 and R of the unobserved reflections was 0.091.

Two types of Fourier map were computed with the final parameters listed in Table 2, one being a difference map and the other a standard electron density map. The difference map showed that all ions were located at positions with nearly zero gradient, and no areas of appreciable electron density remained in the unit cell. As expected, the standard electron density map was little different from the one computed when R was 0.095.

Discussion

Differentiation of the cation positions

As mentioned previously, assignment of the Eu^{2+} and Eu^{3+} positions was based on the cation locations in CaFe_2O_4 . These ions could not be differentiated in the final electron density map, even though ionic scattering factors were used, but their assignments were verified by packing considerations in the final structure. Fig. 1 shows a cross section of the structure at $z=\frac{1}{4}$, with the ions drawn to the same scale as the cell edges. Eu^{2+} and Eu^{3+} radii of 1.17 and 0.95 \AA , respectively, were calculated from the lattice parameters of EuO and cubic-phase Eu_2O_3 (Rau, 1964). These calculations were made assuming an O^{2-} radius of 1.40 \AA (Pauling, 1960), and assuming europium-oxygen contact in the EuO and Eu_2O_3 structures. As shown by Fig. 1, an interchange of the divalent cation, $\text{Eu}(3)$, for either of the trivalent cations, $\text{Eu}(1)$ and $\text{Eu}(2)$, would be impossible without changing some of the positional parameters.

A further check on the cation position assignments was made by means of a cursory X-ray study of the isomorphous compound SrEu_2O_4 (Glasser, Glasser & Schieber, 1965), which contains strontium in place of the divalent europium. A sample of this material produced a powder diffraction pattern which was found to be almost identical with that of Eu_3O_4 , the only notable differences being in the intensities of some of the lines. A set of 32 SrEu_2O_4 powder reflections which could be unequivocally indexed with unique indices was used to carry out three successive structure factor calculations, with the Sr^{2+} ion successively substituted in each of the three possible europium locations. Positional and thermal parameters listed in Table 2 were used for these calculations, and Sr^{2+} scat-

Table 2. *Final parameters for Eu_3O_4*
Eu(1) and Eu(2) are the trivalent ions. Eu(3) is the divalent ion.

	x	$\sigma(x)$	y	$\sigma(y)$	z	B	$\sigma(B)$
Eu(1)	0.4280	0.0001	0.6140	0.0001	0.250	0.476	0.008
Eu(2)	0.4136	0.0001	0.1106	0.0001	0.250	0.466	0.008
Eu(3)	0.2481	0.0001	0.3525	0.0001	0.250	0.594	0.009
O(1)	0.2150	0.0013	0.6824	0.0011	0.250	0.845	0.143
O(2)	0.1334	0.0011	0.9841	0.0009	0.250	0.606	0.111
O(3)	0.0055	0.0014	0.2151	0.0012	0.250	0.947	0.161
O(4)	0.4297	0.0013	0.9207	0.0012	0.250	0.843	0.139

Table 3. Observed and calculated structure factors

Each group of three columns list h, 10F0, and 10Fc respectively. E indicates reflections affected by extinction. * indicates unobserved reflections whose 10F0 are listed as ten times the minimum observable threshold.

Table with multiple columns of numerical data representing structure factors. The table is organized into groups, with each group containing three columns: h, 10F0, and 10Fc. The data is presented in a grid-like format with various sub-headers and numerical values.

tering factors (Cromer & Waber, 1965) were used in place of the Eu^{2+} scattering factors. The best structure factor agreement ($R=0.17$) was obtained with Sr^{2+} in the location assigned to the divalent europium ion, $\text{Eu}(3)$, of the Eu_3O_4 structure. The poorer agreement ($R=0.22$ and $R=0.21$) when Sr^{2+} was assigned to the other two possible europium locations was taken as evidence to rule out those lattice sites for possible occupancy by the divalent cation.

Bonding and coordination

A stereoscopic illustration of the Eu_3O_4 structure, suitable for three-dimensional viewing with a standard 3-D viewer, is shown in Fig. 2. This illustration was prepared by Dr C. K. Johnson of Oak Ridge National Laboratory, using his crystal structure plotting program, ORTEP (Johnson, 1965). The view is along the short c axis, and shows the two atom layers and their relationship to the unit cell edges. Bond lengths shown in the figure were obtained when R was 0.095, and in some cases differ slightly from the final bond lengths.

As may be deduced from Fig. 2, each europium cation is coordinated with oxygen atoms at three levels along the c axis, *i.e.* for europium ions with $z=\frac{1}{4}$, coordinated oxygen atoms are at $z=-\frac{1}{4}$, $\frac{1}{4}$, and $\frac{3}{4}$. The coordination polyhedra are shown in Fig. 3, where the drawings are tilted somewhat about the x axis to separate the projected images of oxygen ions at $z=-\frac{1}{4}$ and $z=\frac{3}{4}$. Bond lengths and angles within these polyhedra, as well as nearest europium–europium distances, are listed in Table 4. Average standard deviations of the distances are 0.001 Å for Eu–Eu, 0.011 Å for Eu–O, and 0.017 Å for O–O, while the deviations of the angles average about 0.3°.

As shown in Fig. 3(a) and (b), the trivalent europium atoms, $\text{Eu}(1)$ and $\text{Eu}(2)$, are coordinated to six oxygen atoms forming distorted octahedra, the $\text{Eu}(2)$ octahedron showing the greatest deviation from regularity. Average $\text{Eu}^{3+}\text{--O}^{2-}$ distances are 2.337 and 2.343 Å, respectively, in the $\text{Eu}(1)$ and $\text{Eu}(2)$ polyhedra, while average $\text{O}^{2-}\text{--O}^{2-}$ distances are 3.295 Å in the former

and 3.280 Å in the latter. Both the shortest (2.955 Å) and the longest (3.892 Å) significant oxygen–oxygen distances in Eu_3O_4 occur in the $\text{Eu}(2)$ polyhedron. The average bond angle involving two adjacent oxygen atoms and europium is 89.9° in the $\text{Eu}(1)$ polyhedron and 89.5° in the $\text{Eu}(2)$ polyhedron, both very close to the undistorted octahedral value of 90°.

In the divalent $\text{Eu}(3)$ polyhedron [Fig. 3(c)] the europium ion is located in the center of a triangular prism whose interfacial angles are 49.8°, 62.9°, and 67.3°. Six oxygen ions, three at $z=-\frac{1}{4}$ and three at $z=\frac{3}{4}$, make up this prism, and two additional oxygen ions, located out from the two wide faces of the prism on the $z=\frac{1}{4}$

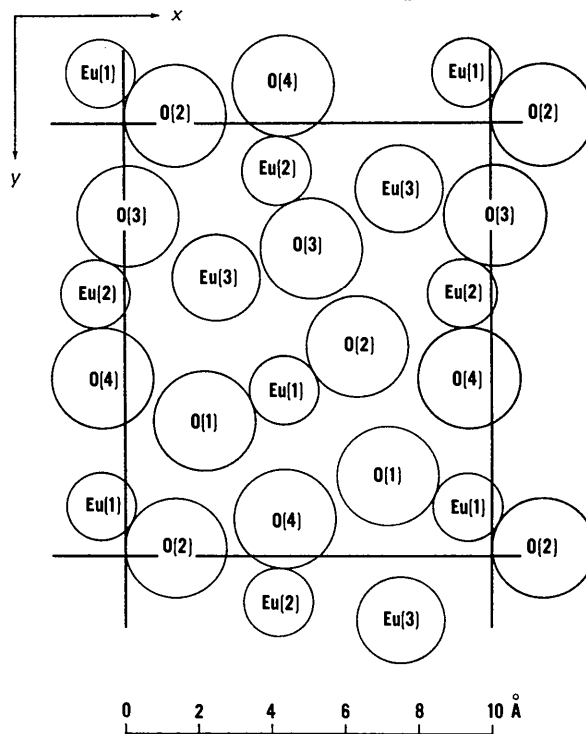


Fig. 1. Cross section of the Eu_3O_4 structure at $z=\frac{1}{4}$. The ions are drawn to the same scale as the cell edges.

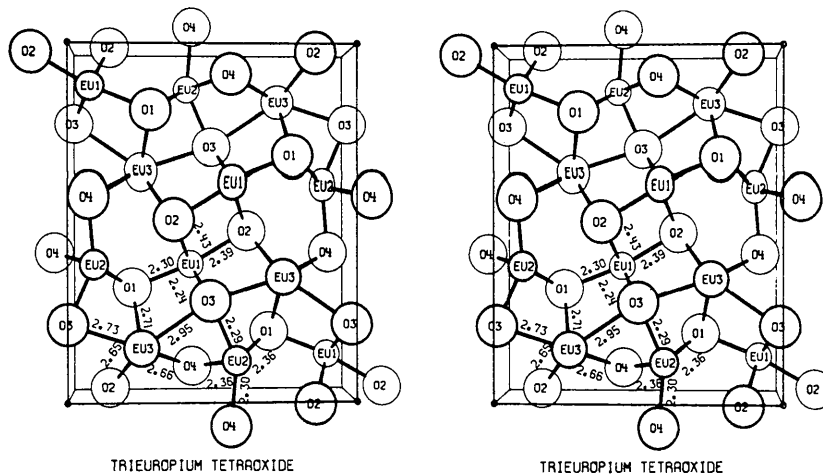


Fig. 2. Stereoscopic illustration of the Eu_3O_4 structure viewed along the c axis. The box represents one unit cell.

Table 4. Bond lengths and angles in Eu_3O_4

Primed ions have $z = \frac{1}{2}$ or $z = -\frac{1}{2}$, unprimed ions have $z = \frac{1}{2}$. The second column in each grouping indicates the bond or angle multiplicity. Average standard deviations of the distances are 0.001 Å for Eu–Eu, 0.011 Å for Eu–O, and 0.017 Å for O–O; the deviations of the angles average about 0.3°.

Within the trivalent Eu(1) polyhedron

Eu(1)–O(1)	1	2.301 Å	O(1)–Eu(1)–O(2)	1	171.3°
Eu(1)–O(2)	1	2.386	O(2')–Eu(1)–O(3')	2	172.8
Eu(1)–O(2')	2	2.430	O(1)–Eu(1)–O(2')	2	89.6
Eu(1)–O(3')	2	2.236	O(1)–Eu(1)–O(3')	2	94.9
O(1)–O(2')	2	3.334	O(2)–Eu(1)–O(2')	2	84.4
O(1)–O(3')	2	3.342	O(2)–Eu(1)–O(3')	2	90.5
O(2)–O(2')	2	3.234	O(2')–Eu(1)–O(2')	1	92.2
O(2)–O(3')	2	3.285	O(2')–Eu(1)–O(3')	2	82.1
O(2')–O(2')	1	3.502	O(3')–Eu(1)–O(3')	1	103.1
O(2')–O(3')	2	3.069			
O(3')–O(3')	1	3.502			

Within the trivalent Eu(2) polyhedron

Eu(2)–O(3)	1	2.297 Å	O(3)–Eu(2)–O(4)	1	152.2°
Eu(2)–O(4)	1	2.295	O(1')–Eu(2)–O(4')	2	166.3
Eu(2)–O(1')	2	2.345	O(3)–Eu(2)–O(1')	2	83.4
Eu(2)–O(4')	2	2.388	O(3)–Eu(2)–O(4')	2	83.0
O(3)–O(1')	2	3.088	O(4)–Eu(2)–O(1')	2	114.0
O(3)–O(4')	2	3.104	O(4)–Eu(2)–O(4')	2	78.2
O(4)–O(1')	2	3.892	O(1')–Eu(2)–O(1')	1	96.6
O(4)–O(4')	2	2.955	O(1')–Eu(2)–O(4')	2	82.9
O(1')–O(1')	1	3.502	O(4')–Eu(2)–O(4')	1	94.3
O(1')–O(4')	2	3.136			
O(4')–O(4')	1	3.502			

Within the divalent Eu(3) polyhedron

Eu(3)–O(4')	2	2.638 Å	O(3)–Eu(3)–O(3)	1	128.5°
Eu(3)–O(3)	1	2.955	O(3)–Eu(3)–O(1)*	1	112.2
Eu(3)–O(1')	2	2.722	O(1)–Eu(3)–O(3)*	1	119.3
Eu(3)–O(3)	1	2.720	O(3)–Eu(3)–O(4')	2	67.2
Eu(3)–O(2')	2	2.648	O(3)–Eu(3)–O(1')	2	72.0
Eu(3)–O(1)*	1	3.991	O(3)–Eu(3)–O(1')	2	69.2
O(3)–O(4')	2	3.104	O(3)–Eu(3)–O(2')	2	75.5
O(3)–O(1')	2	3.342	O(4')–Eu(3)–O(1')	2	84.3
O(3)–O(1')	2	3.088	O(1')–Eu(3)–O(1')	1	80.1
O(3)–O(2')	2	3.285	O(4')–Eu(3)–O(4')	1	83.2
O(4')–O(4')	1	3.502	O(1')–Eu(3)–O(2')	2	87.9
O(4')–O(1')	2	3.597	O(2')–Eu(3)–O(2')	1	82.8
O(1')–O(1')	1	3.502	O(2')–Eu(3)–O(4')	2	71.4
O(1')–O(2')	2	3.728	O(4')–O(1')–O(2')	2	49.8
O(2')–O(2')	1	3.502	O(1')–O(2')–O(4')	2	62.9
O(2')–O(4')	2	3.084	O(2')–O(4')–O(1')	2	67.3

* Quantities involving the ninth 'non-coordinated' oxygen ion.

Nearest europium-europium distances

Eu(1)–Eu(3)	1	3.638 Å
Eu(1)–Eu(1')	2	3.568
Eu(1)–Eu(3')	2	3.728
Eu(1)–Eu(2')	2	4.079
Eu(1)–Eu(3')	2	3.805
Eu(1)–Eu(2')	2	3.865
Eu(1)–Eu(1)	2	3.502
Eu(2)–Eu(3)	1	3.360
Eu(2)–Eu(1')	2	3.865
Eu(2)–Eu(3')	2	3.925
Eu(2)–Eu(2')	2	3.634
Eu(2)–Eu(3)	1	3.403
Eu(2)–Eu(1')	2	4.079
Eu(2)–Eu(2)	2	3.502
Eu(3)–Eu(3)	1	3.403
Eu(3)–Eu(1')	2	3.805
Eu(3)–Eu(2)	1	3.360
Eu(3)–Eu(1')	2	3.728
Eu(3)–Eu(1)	1	3.638
Eu(3)–Eu(2')	2	3.925
Eu(3)–Eu(3)	2	3.502

level, complete the eightfold coordination. A ninth oxygen ion, located out from the third prism face on the $z = \frac{1}{4}$ level, is 3.991 Å from the central europium ion and is probably too distant to be considered in the primary coordination shell. This ninth oxygen ion is indicated by the dashed bond in Fig. 3(c), and the few quantities involving it are marked with an asterisk in Table 4. The average $\text{Eu}^{2+}-\text{O}^{2-}$ distance, omitting this ninth oxygen ion, is 2.711 Å, and the average $\text{O}^{2-}-\text{O}^{2-}$ distance is 3.351 Å, both slightly greater than similar distances in the trivalent europium polyhedra.

The metal ions form a pseudo-hexagonal network, with half the europium ions on $z = \frac{1}{4}$ levels and half on $z = \frac{3}{4}$ levels. Each Eu(1) ion has 13 nearest europium neighbors at an average distance of 3.749 Å, each Eu(2) ion has 12 nearest europium neighbors at an average distance of 3.731 Å, and each Eu(3) ion has 11 nearest europium neighbors at an average distance of 3.666 Å.

Most MN_2O_4 compounds (where M is a divalent metal and N is a trivalent metal) crystallize in structures which are characterized by close-packing of the anions (*i.e.* spinel structures, *etc.*). In the case of Eu_3O_4 the cations are too large to be accommodated interstitially in a close-packed oxygen lattice, and cation size therefore plays an important role in determining the structure. If cation/anion radius ratios are considered, that of $\text{Eu}^{3+}/\text{O}^{2-}$ (ratio = 0.68) predicts sixfold coordination, and that of $\text{Eu}^{2+}/\text{O}^{2-}$ (ratio = 0.84) predicts eight- or ninefold coordination (Pauling, 1960). The fact that these predicted coordinations coincide with those observed for the two types of europium in Eu_3O_4 further tends to confirm the assignment of cation locations in the structure.

Other similar structures

The structural similarity of Eu_3O_4 and CaFe_2O_4 has already been pointed out. It should be mentioned that

$\beta\text{-CaCr}_2\text{O}_4$ (Hill, Peiser & Rait, 1956) and PbBi_2S_4 (Wickman, 1951) are also similar. However, in the reported structure analysis of the latter compound the cation locations were probably incorrectly assigned. In particular, it seems unlikely that one of the trivalent bismuth ions should occupy the position of eightfold coordination, while the other trivalent bismuth ion and the divalent lead ion occupy the two octahedral sites.

An interesting variation of the Eu_3O_4 structure is exhibited by Rh_4P_3 (Rundqvist & Hede, 1960). In this compound, half the atoms are moved one half of a translation distance along the short axis of the unit cell, resulting in a structure having each phosphorus atom at the center of a trigonal prism of rhodium atoms.

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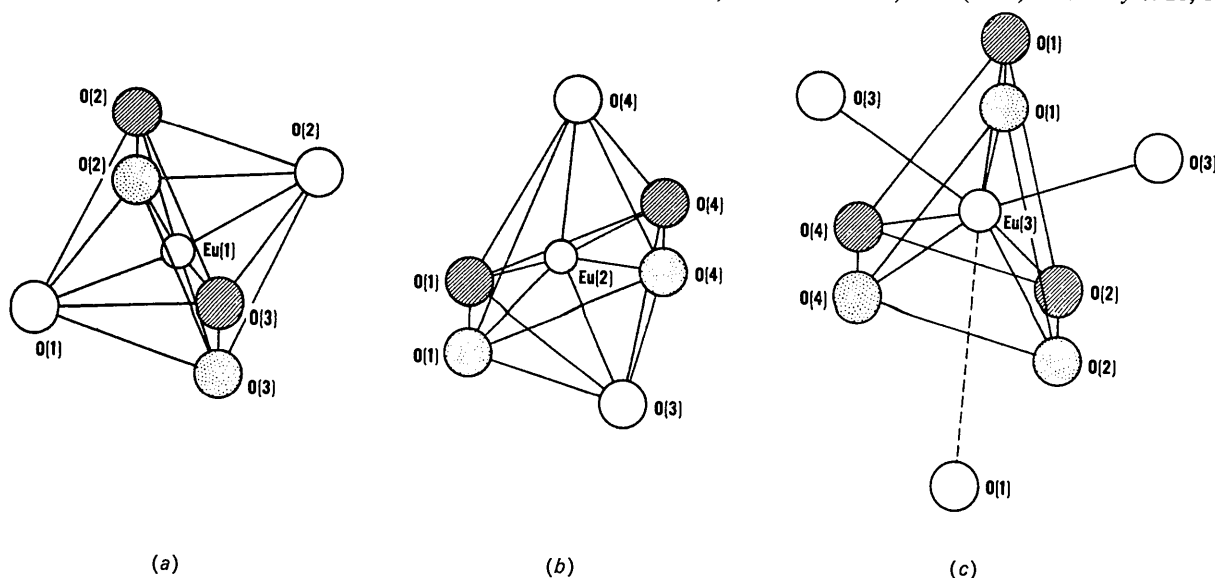


Fig. 3. Europium coordination polyhedra in Eu_3O_4 . Open circles have $z = \frac{1}{4}$, dashed circles have $z = -\frac{1}{4}$, and stippled circles have $z = \frac{3}{4}$. (a) Trivalent Eu(1) polyhedron. (b) Trivalent Eu(2) polyhedron. (c) Divalent Eu(3) polyhedron.

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Refinement of the Crystal Structure of In_2O_3 at two Wavelengths

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Indium sesquioxide is cubic with 16 molecules per unit cell ($a = 10.117 \pm 0.001$), space group $Ia\bar{3}$. The crystal structure has been refined from single-crystal X-ray diffraction data with a precision corresponding to an R index of 0.03. The interatomic distances have been determined to an accuracy of 0.01 Å for the In–O distances and 0.02 Å for the O–O distances. The In–O bond lengths are: (6) In(1)–O 2.18 Å, (2) In(2)–O 2.13 Å, (2) In(2)–O 2.19 Å, (2) In(2)–O 2.23 Å. The large difference in the In(2)–O bond lengths is attributed to an unequal distribution of the repulsive forces among the oxygen atoms that form the polyhedron around In(2).

Since the absorption correction was rather high, the intensity data were taken with two radiations, Mo $K\alpha$ and Cu $K\alpha$, and two independent refinements were made and compared. Contrary to what one would expect, the extinction correction is much larger for the Mo data than for the Cu data.

The possible existence of a high pressure phase of In_2O_3 is discussed.

Introduction

Many of the metallic sesquioxides, M_2O_3 , crystallize in the corundum type and the A , B and C rare-earth sesquioxide structures. The corundum type and the A modification are trigonal, space group symmetry $R\bar{3}c$ and $P\bar{3}m1$ respectively, the B modification is monoclinic, $C2/m$, and the C modification is cubic, $Ia\bar{3}$. This paper reports the refinement of the crystal structure of indium sesquioxide which crystallizes in the cubic C modification (Zachariasen, 1927). Many other sesquioxides crystallize with the same structure, for instance: Sc_2O_3 , Y_2O_3 , Tl_2O_3 , Pu_2O_3 , Cm_2O_3 , Am_2O_3 , R_2O_3 (R =rare earth) *etc.* Some of these oxides can also exist in the A and/or B forms. The atomic arrangement of the C modification was first determined for the mineral bixbyite $(\text{Fe,Mn})_2\text{O}_3$ (Pauling & Shappell, 1930). In 1956 Dachs refined this structure and gave a more detailed description of it. Since then neutron diffraction studies using powders have been made on several of the C type rare-earth sesquioxides (Table 5).

Recently Remeika & Spencer (1964) of this laboratory prepared single crystals of In_2O_3 by a flux technique for electrical conductivity measurements. These crystals have the shape of rectangular prisms and are

dark green. Since single crystals of In_2O_3 were available, an accurate determination of a C modification structure was undertaken with the use of modern techniques to measure X-ray intensities. This refinement provides accurate values of the In–O distance which are necessary for the theoretical and experimental studies on In_2O_3 being carried out at present in this laboratory.

Crystallographic data

Crystals of In_2O_3 are body-centered cubic with 16 molecular weights (M.W. 277.64) in a cell of dimension $a = 10.117 \pm 0.001$ Å (Roth, 1957), the calculated density is 7.120 g.cm^{-3} ; hkl is present only with $h+k+l=2n$ and $0kl$ only with $k,l=2n$; the space group is hence taken as $Ia\bar{3} (T_h^7)$. The 32 metallic atoms are in the special positions, 24 in (d) and 8 in (b), while the 48 oxygen atoms are in the general position.

Experimental

All intensity measurements were made with a General Electric XRD-5 spectrometer equipped with a single-crystal orienter and a scintillation counter. The integrated intensities were obtained manually with the